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AD0487560

AFML-TR-65-214

# CENTER NOTCH PLANE STRAIN K, FRACTURE TOUGHNESS PROPERTIES OF SEVERAL HIGH-STRENGTH STEEL ALLOYS

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TECHNICAL REPORT AFML-TR-65-214

OCTOBER 1965

AIR FORCE MATERIALS LABORATORY
RESEARCH AND TECHNOLOGY DIVISION
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WRIGHT-PATTERSON AIR FORCE BASE, OHIO

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#### FOREWORD

This report was prepared by Mr. Sidney O. Davis, Nathan G. Tupper, 1/Lt, USAF, and D. C. LaGrone, 1/Lt, USAF, Materials Information Branch, Materials Applications Division, Air Force Materials Laboratory, Research and Technology Division, Air Force Systems Command, Wright-Patterson Air Force Base, Ohio, and Roger M. Niemi of Monsanto Research Corporation, Dayton, Ohio. This program was conducted under Project No. 7381 "Materials Applications," Task No. 738106, "Design Information Development."

This report covers work conducted from January 1964 to January 1965. The manuscript was released by the authors in June 1965 for publication as an RTD Technical Report.

The authors express their appreciation to Mr. Julius Teres, Air Force Materials Laboratory, for his support and assistance in designing the Acoustical Indicator Electronic Circuit.

This technical report has been reviewed and is approved.

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Chief, Materials Information Branch Materials Applications Division Air Force Materials Laboratory

### ABSTRACT

Plane strain fracture toughness and tensile properties were determined at room temperature, utilizing compliance and pop-in methods, for four high-strength steel alloys: PH 15-7 Mo, 17-7 PH, AM 350 and Vasco Jet 1000 (H-11). Fracture toughness values varied over a fairly wide range, with AM 350 having the highest at approximately 60.7 KSI √in. and Vasco Jet 1000 (H-11) having the lowest at approximately 25 KSI √in. A computer program used to reduce fracture toughness data was able to calculate critical crack length as well as fracture toughness when given either suitable compliance gage data or the measured test data. An acoustical pickup, used as an additional test monitor, is described. Analytical basis for the compliance method is presented.

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## INTRODUCTION

As the quest for better performance aerospace systems continues, material strength requirements become higher and higher. The use of impact energy absorption test data has become inadequate for describing the toughness characteristics of materials. In the continuing search for high-strength material, it becomes more and more important to know (with accuracy) the behavior of a material (as a formed aerospace component) under service conditions. Therefore, a proposed material must be evaluated for its resistance to brittle fracture in the thicknesses, strength levels, and metallurgical conditions envisioned. If such an evaluation is not made, the behavior of the material is not completely characterized and either overdesign or underdesign can result. Both cases are undesirable -- overdesign will result in a heavy, inefficient structure, while underdesign can result in catastrophic failure.

The plane strain fracture toughness parameter,  $K_{\rm IC}$ , gives to the designer some measure of service performance when a small flaw is present in the material. By knowing the size of the largest flaw which can be tolerated in a structure, the designer can determine the stress level which will cause failure, and then design with an appropriate safety factor. Alternatively, he can determine the smallest flaw detectable with available inspection equipment, determine the stress level which will cause failure with this flaw present, and design below that stress.

Four high-strength steel alloys in sheet form were tested in this program. This report shows that the plane strain fracture toughness of high-strength alloys is very important even for relatively thin sheet.

#### MATERIALS

Table 1 shows a list of the four sheet materials, their thicknesses, and heat-treatment conditions used in this program. Standard heat-treatment designations were: PH 15-7 Mo, RH 950; 17-7 PH, RH 1050; AM 350, SCT 950; and Vasco Jet (VJ) 1000 (modified AISI type H-11), triple tempered at 1000°F. The alloy suppliers, heat numbers, and the chemical composition of the materials used in this program are also listed. Photomicrographs of the four sheet materials in the heat-treated condition are shown in Figure 1.

#### SPECIMENS

The fracture toughness tests conducted in this program used a 3 in. by 12 in. center notch specimen. A drawing of this specimen is shown in Figure 2.

The specimen preparation procedure was as follows: shear to size, drill 1 in. dia pinholes, heat-treat in fixtures, surface grind to remove scale, "Elox" center notch, drill shim holes, and tension-tension fatigue crack the starter notch to approximately 1 in. (1/3 W). The center notch was placed in the specimen after heat treatment to avoid any heat-treatment distortion problems. The specimens were heat-treated in an atmosphere of dissociated ammonia.

Recently, the ASTM committee on Fracture Testing of High-Strength Metallic Materials recommended that the specimen width-to-thickness ratio should be between 16 and 45. However, the specimens for this program were machined prior to this recommendation and the width-to-thickness ratios ranged from 50 for VJ 1000 to 85 for 17-7 PH. Because of the relatively thin sheet material used, considerable difficulty was experienced with buckling around the pinholes. To prevent this buckling, hardened steel carrier shims were used. These shims

TABLE 1. MATERIALS DATA

OESIGNATION NOITANDISTO					
SU JAS	RH 950	RH 1050	SCT 950	1	
MANUFACTURERS' HEAT NUMBERS	1	T56038 RH	89703	1	
INJ SPEC NO.	Soln: 1750° F-15 min Air Cool -100° F-8 hrs Age: 950° F-1 hr	Soln: 1750° F-15 min Air Cool -100° F-8 hrs Age: 1050° F-1 hr	Soln: 1710° F-1 hr Air Cool -100° F-5 hrs Age: 950° F-3 hrs	Soln: 1850° F-1 hr Air Cool Age: 1000° F (2+2+2 hrs)	
		25043	AMS 5548	l	
10.0	0,053	0.038	0.052	0,062	
NOI TIONO SA	A	A	н	I	
NOITISOGMOS NOITISOGMOS	Cr-15.32, Ni-7.11, Mo-2.39, Si-0.26, C09, A1-1.2 Mn55, Fe-bal	Cr-17.18, Ni-7.16 A1-1.20, Mn66, Si23, C09, P018, S008	Cr-16.42, Ni-4.3, Mo-2.88, Si27, C092, P016, S018, Co035, N10, Cu12 A1047, Fe-bal	Cr-5.34, Mo-1.36, Si87, C39, Mn24, V48	
THICKNESS (IM.)  MANUFACTURER		ARMCO	0.050 Alleghany Ludlum	0.060 Vanadium Alloys	
THICKNE	0.050	0.036	0.050	0.060	Annealed Heat treated
INIE RIAL	PH 15-7 Mo 0.050 ARMCO	17-7 PH	AM 350	VASCO JET 1000	* A - Annealed H - Heat trea

were fastened to the specimen by pins located in the three peripheral holes shown in Figure 2. These carrier shims, while not fastened to the grips, created more load bearing surface and distributed the stress around the pinholes. This method of shimming, as described in the first ASTM Fracture Toughness of High-Strength Steel Committee Report (Reference 1) offers a fast and convenient means to eliminate pinhole buckling problems.

Standard 2-in. gage length smooth tensile specimens were machined and heat-treated simultaneously with the fracture toughness specimens.

# TEST EQUIPMENT

All tensile and fracture-toughness testing was done on a 50,000 lb Baldwin Universal tensile machine. The fracture specimens were fatigued-cracked in tension-tension using a 6-ton Schenck fatigue machine shown in Figure 3. Fatiguing at a maximum stress of 1/5 the yield stress at 2000 cycles/min produced fatigue cracks of the desired length in approximately 15 minutes. The brittle behavior of VJ 1000 necessitated a lowering of the maximum stress after crack initiation to prevent the fatigue crack from becoming unstable as it approached the desired length of 1 in. Fatigue crack lengths were measured after testing with a toolmaker's microscope.

The compliance gage chosen was originally developed by Richard W. Boyle at the Naval Research Laboratory (Reference 2). This compliance gage was modified to use a Model PS-6M Baldwin Microformer which provides a total magnification of approximately 550X. The compliance record was made on a Baldwin autographic recorder.

During the program it was found that the magnification produced by this combination of microformer and gage was insufficient to reliably record the pop-in event. It was then decided to obtain an acoustical indication of crack growth during the fracture test to help define the compliance pop-in  $(K_{\underline{IC}})$  event. An uncalibrated acoustical indication was recorded on the

Y axis simultaneously with load on the X axis of a Houston model HR-97 X-Y recorder. The load signal was obtained by attaching an auxiliary slide wire to the load indicator of the 50,000-lb tensile machine. The slide wire was one leg of a Wheatstone bridge measuring circuit. This gave an X axis record of the bridge unbalance as indicated by the position of the wiper arm on the slide wire. The bridge output was calibrated with the tensile machine load indication. An RCA magnetic cartridge (1940 vintage) with a 0.050 inch diameter wire pickup arm inserted in the notch gave an indication of crack activity. The acoustical signal was amplified with a Ballantine model 300 electronic voltmeter, filtered with a 20 KC high pass filter to eliminate machine noise, and rectified for the Y axis input. An additional monitor was obtained by using a high impedance Brush crystal head set in parallel to the X-Y recorder. A schematic of the acoustical monitoring circuit is shown in Figure 4.

Pictures of the test setup and a close-up of the compliance gage attached to the specimen are shown in Figures 5 and 6.

## PROCEDURE

Wherever applicable, test procedures used were those recommended by the ASTM Committee on Fracture Testing of High-Strength Metallic Materials (References 3, 4 and 5).

After the specimens were fatigue-cracked, it was necessary to heat-tint the fatigue crack surface to make the starter crack visible for optical measurement. Heat-tinting was done at approximately 600°F for one hour.

The test data were obtained from the compliance gage recordings. The acoustical indications, previously described, were used to help determine the pop-in load when no distinct pop-in occurred on the compliance record. In some cases, the deviation from linearity of the compliance slope was used as a criteria for pop-in load determination. The lack of adequate elastic constraint in the sheet materials contributed to the failure to obtain distinct pop-in load values. The fracture-toughness value obtained by using the deviation from linearity is commonly designated by  $K_{nc}$  and has been used by other investigators when no clearly defined pop-in has occurred. In the tabulated data, a distinction has been made between  $K_{lc}$  and  $K_{nc}$ . For some of the materials, the acoustical indications helped to interpret the data; however, the use of acoustical information has not been formally endorsed by the ASTM Committee.

A thin sheet center notch specimen has a predominate plane stress state rather than a plane strain state required for  $K_{IC}$  conditions. For this reason the center notch fracture-toughness specimen is not considered an optimum specimen for reproducible  $K_{IC}$  type measurements. Therefore, the  $K_{IC}$  data presented in this report are primarily indicative only of the fracture initiation resistance properties of the materials.

Since it was impossible to prevent the compliance gage knife edges from moving during the latter stages of crack propagation, optical measurement of the critical crack length was not conducted. Instead, the sharp-notch strength has been used to indicate the fracture resistance of the material. The sharp-notch strength of a material is defined as the maximum load sustained before fracture divided by the nominal cross-sectional area, i.e.,

where  $P_{\max}$  is the maximum load sustained, W is the width, B is the thickness and  $2a_0$  is the original crack length.

### COMPLIANCE GAGE CALIBRATION

To verify the compliance calibration curve reported by Richard W. Boyle (Reference 2), a gage calibration test for center notch specimens was conducted as a part of this program.

Nine 3 in. by 12 in. center-notch specimens of AM 350 were tested. Simulated cracks were prepared by hand sawing slot lengths ranging from 0 to 1.6 inches.

In this method of compliance gage calibration a plot of  $Ev/\sigma W$  (ordinate) vs.  $\pi a/W$  (abscissa) is used to calculate the crack length (2a) when the modulus of elasticity (E), specimen dimensions, and the deflections at all loads are known. Interested readers may refer to Mr. Boyle's discussion (Reference 2) for the mechanics of using the calibration curve.

Figure 7 compares our calibration curve with R. W. Boyle's curve. The excellent correspondence of curve shape is shown. The difference of ordinate (Ev/ $\sigma$ W) values is accounted for by the difference in Modulus of Elasticity of the materials used in the tests, i.e., steel versus aluminum, respectively.

The compliance gage calibration curve method for calculating the critical crack length (2a) of a material has been programmed as an optional part of the computer program which is discussed later.

The Irwin-Westergaard analytical basis for compliance measurements of fracture toughness is presented in Appendix II.

# RESULTS AND DISCUSSION

Table 2 shows the base-line tensile test values which characterize the material used in the fracture-toughness tests. The yield strength value for PH 15-7 Mo is an estimated value obtained from existing literature. As discussed in Reference 1, an error of 10 percent in the yield stress will result in a 2 percent error in the fracture-toughness value when corrected for the plastic zone size.

A complete tabulation of the fracture data obtained for the four materials can be found in Tables 3, 4, 5 and 6. The initial stress intensity value of K has been tabulated as K because the majority of K measurements were based on the deviation from linearity of the compliance gage record. Where a distinct pop-in occurred it is noted by an asterisk in the data tables.

All the materials tested had low notch strength to yield strength ratios. The following tabulation lists the fracture-toughness values obtained. Generally, the longitudinal specimen values were higher than the transverse values. A longitudinal specimen has its major axis parallel to the rolling direction of the material.

		K	Notch Streng	th Ratio
	Longitudinal	<u>nc</u> Transverse	Longitudinal	Transverse
PH 15-7 Mo	41.3	40.4	. 241	. 264
17-7 PH	52.6	50.3	.798	. 658
AM 350	60.7	<b>54.</b> 1	. 972	. 891
VJ 1000 (H-11)	25.3	25.0	.142	. 135

Rockwell C hardness readings of the specimens tested showed consistent hardness values over the entire specimen. The Rc hardness ranges were: PH 15-7 Mo, 49.7 to 50.2; 17-7 PH, 42.5 to 43.8; AM 350, 44.5 to 45.5; VJ 1000, 58.2 to 59.5.

The fracture test values obtained were relatively consistent with the macroscopic fracture surface observations. A fracture surface with granular appearance corresponds to cleavage, or brittle fracture, while a fibrous appearance corresponds to a ductile failure.

	% Sh	ear	Appearan	ce of	Fracture
D.T. 4.5 3.4.	Long.	Trans.	Longitudinal		Transverse
PH 15-7 Mo	5	5	Rough	and	Granular (G)
17-7 PH	85-90	30-40	Fibrous (F)	-	Mixed F & G
AM 350	95-100	40-50	Fibrous	-	Mixed F & G
VJ 1000	5	5	Smooth	and	Granular

The difficulties experienced in interpretation of some of the compliance curves have shown that a higher magnification than the 550X extensometer used in this program is desirable.

Jones and Brown (Reference 6) have recently reviewed the use of acoustical crack growth detection procedures and their value to the investigator. Their equipment has greater sensitivity and sophistication than the circuit described in Figure 4; however, some of their observations have been substantiated by this program. In particular, as the specimen thickness

decreases, the signal-to-noise ratio decreases, thus permitting extraneous acoustical indications prior to a "pop-in" or a deviation from linearity. Extraneous noise is a problem especially with a screw driven test machine. The acoustical pickup serves to give the investigator an additional monitor during a test and may help to determine the "pop-in" load of the material.

The only comparative data values found in the literature were for VJ 1000 in Reference 4. A mean value of approximately 31 KSI  $\sqrt{\text{in}}$ . for  $K_c$  found therein compares favorably with the  $K_{Ic}$  value of 25 KSI  $\sqrt{\text{in}}$ . and the  $K_c$  values from 30 to 33 KSI  $\sqrt{\text{in}}$ . obtained from the compliance records in this program.

## CONCLUSIONS

- 1. This work has shown that of the alloys tested, VJ 1000 (H-11) has the lowest fracture toughness while AM 350 has the highest, although all values were lower than expected for the particular heat treatments tested.
- 2. The longitudinal and transverse fracture-toughness values were very similar, although the longitudinal values were slightly higher for the materials tested.
- 3. An acoustical monitor is of value in determining the pop-in load where a definite step pop-in criterion is used.
- 4. Fracture-toughness values obtained using the load at deviation from linearity of the compliance record did not differ appreciably from those obtained using a pop-in load.
- 5. A computer program greatly simplifies reduction of fracture-toughness data, especially when the compliance method is used.
- 6. The center-notched, fracture-toughness specimen is not a practical specimen for determination of plane strain fracture toughness. However, it is highly recommended as a plane stress specimen.

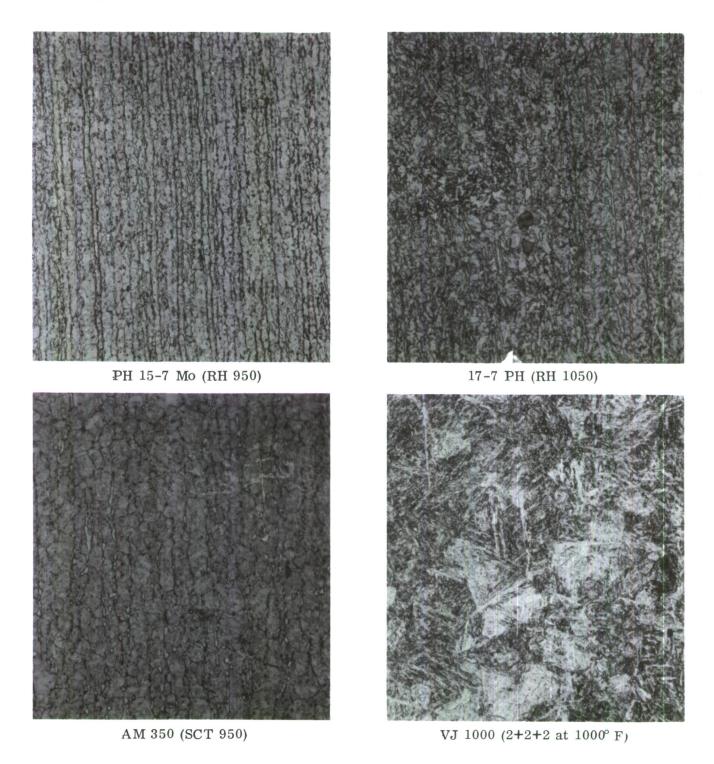


Figure 1. 500X Photomicrographs of Materials Tested. PH 15-7 Mo, 17-7 PH, AM 350 were etched with a solution of 5g FeCl $_3$ , 50 ml HCl, and 100 ml H $_2$ 0, while VJ 1000 was etched with 4g C $_6$ H $_2$ (NO $_2$ ) $_3$ OH and 100 ml CH $_3$ OH $_1$ .

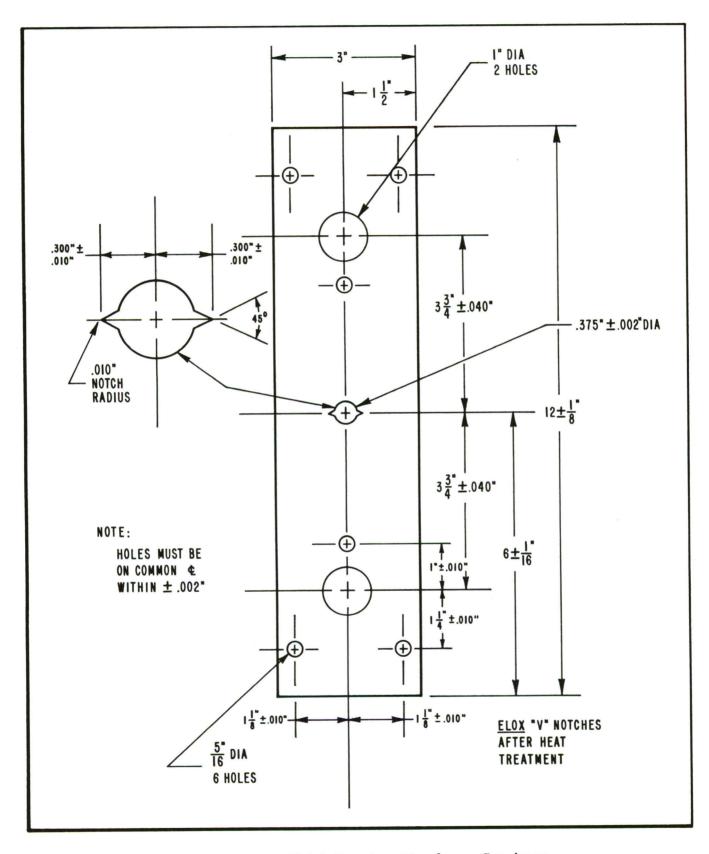


Figure 2. Center Notch Fracture Toughness Specimen

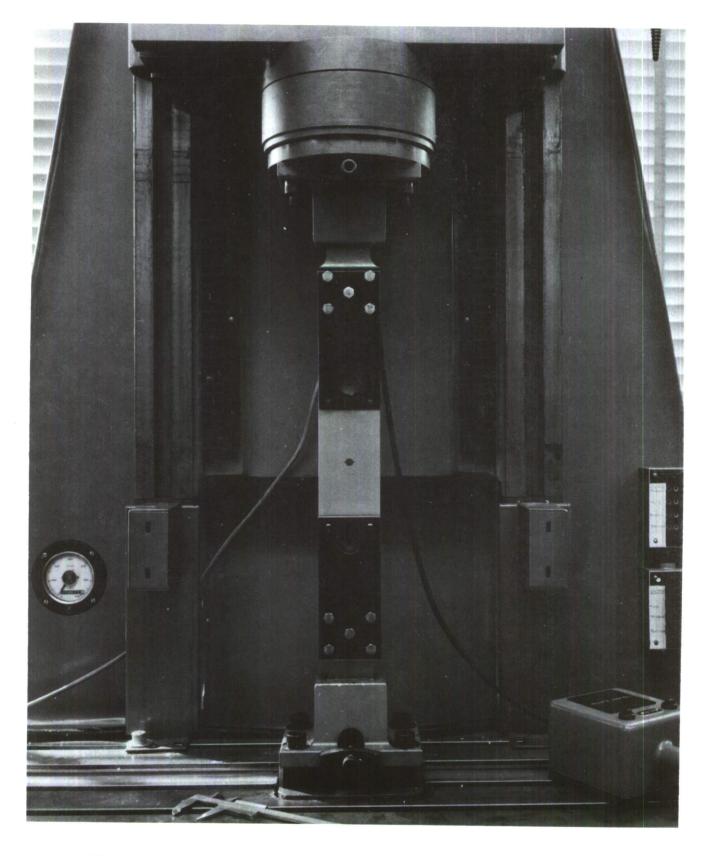


Figure 3. Fatigue Cracking Setup on a Six-Ton Schenck Fatigue Machine

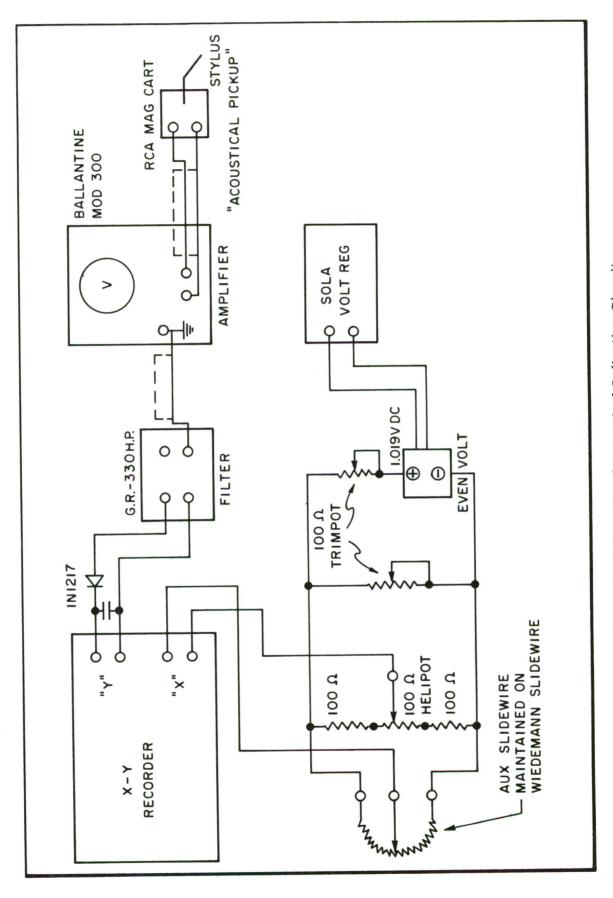


Figure 4. Schematic of Acoustical Indication Circuit

Figure 5. Test Equipment

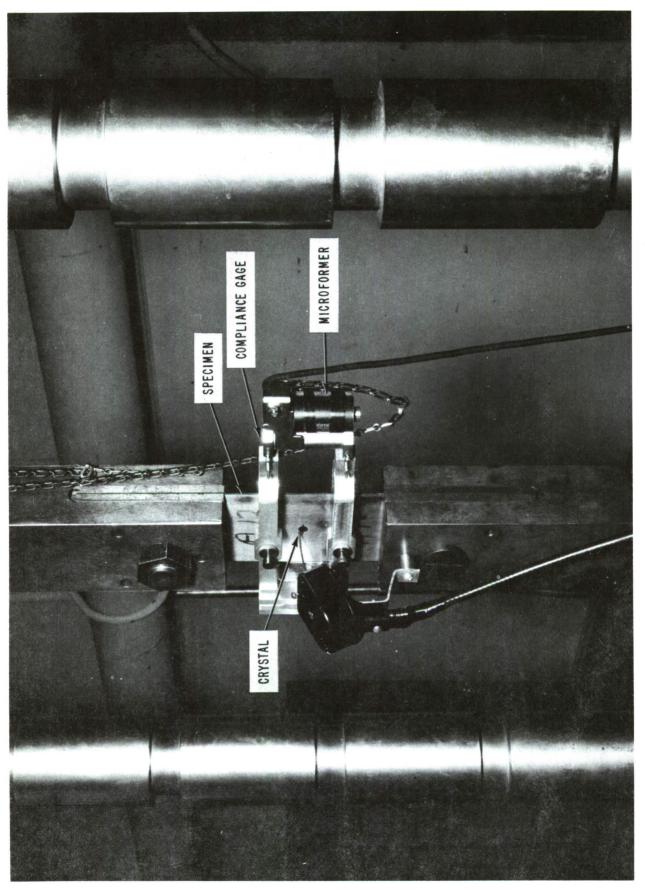


Figure 6. Close-up of Compliance Gage and Acoustical Pickup

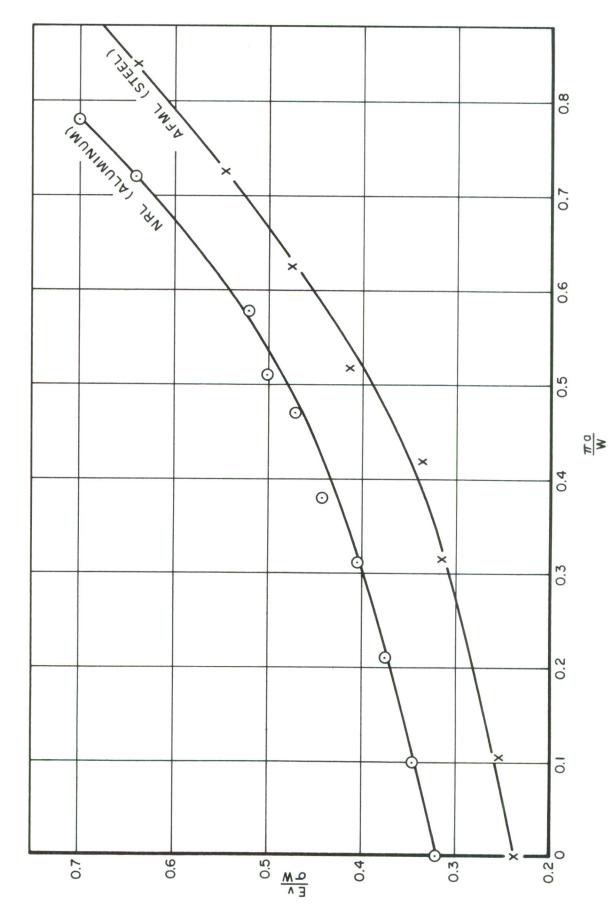


Figure 7. Comparison Compliance Gage Calibration Curves

TABLE 2. TENSILE TEST RESULTS

MATERIAL	SPECIMEN	24 VIELO STAFSS (4.	ULTIMATE STRES	% ELONGATION 11 2 11.
17-7 PH	L	185.3	191.6	3.2
	L	188.0	194.8	4.0
	T	184.5	191.2	5.3
	T	186.3	190.4	5.0
AM 350	L	170.8	192.4	10.8
	L	169.4	194.1	10.5
	T	161.6	188.8	13.8
	T	165.0	189.0	13.8
VJ 1000 (H-11)	L	252.9	308.0	5.3
	L	248.4	301.3	5.2
	T	236.1	296.3	4.8
	T	234.7	297.5	4.8
PH 15-7 Mo*	L	220. 0	240.0	8.0
	T	225. 0	250.0	10.0

<sup>\*</sup> Estimated Values from "Air Weapons Materials Application Handbook" (ARDC TR 59-66)

TABLE 3. FRACTURE TEST RESULTS FOR PH 15-7 Mo

NI-dOd LONILSIO											
(A) LEN ON TOWN	1	!	!			!	!	+			٦
1) 201	42.6	41.1	39.5	1 1			40.9	42.4			
(U) 0,	900.	900.	. 005	1 1	1 1	1 1	.005	900'			
* 0110 #10N3 * * 011 PA		.24	.25	. 22	24	.24	.26	.26			
* 011AA TAN	. 22	.21	.20				.21	. 22			
13/1	220.0	220.0	220.0	220.0	220.0	220.0	220.0	220.0			
SSJATS (18N) SSJATS (18N)	48.0	46.3	44.6			1 1	46.4	47.6			
SSJATS (ISN) SSJATS (ISN) SSJATS (ISN)	48.0	46.3 53.6	44.6 54.9	49.5	51.7	52.6	46.4	47.6			
049	31.5	30.5	29.5 36.4	31.7	33.3	36.6	31.3	31.2	*		in
SANCTURE LOAD		5.33	5.48	4.97	5.02	5.52	5.86	5.62	d stress	stress	a distinct pop-in
UNOT CHACK  ONOT ON  ONOT ON  ONOT ON  ON  ON  ON  ON  ON  ON  ON  ON  ON	4.75	4.60	4.45		1 1		4.70	4.74	at P <sub>nc</sub> /yield	ss/yield	of a disti
(NI) SOUTH CANAL	1.038	1,029	1.017	1,082	1,069	.921	976.	1.040		ire stre	rrence o
THICKNESS (IN)	. 050	020.	. 050	. 052	. 0502	. 0500	. 050	. 0503	Net section stress	Nominal fracture stress/yield	Indicates occurrence of
NOITO3A10	3.02	3.02	3.01	3.01	3.00	3.02	3,00	3.02	Net se		Indicat
3810	L	H	I	I	IJ	IJ	ı	I	*	* * 2	+-

TABLE 3. FRACTURE TEST RESULTS FOR PH 15-7 Mo (cont.)

NI-904 TONITSIO								
(W) ISIN	+	+	1	*	1		1	<u> </u>
Ans (No	40.9	39.4	11	38.5	42.8	40.6		
(4/9, ** 0/1)	. 005	. 005	1 1	. 005	900.	. 005	11	
* 011N # 011N # 13N \$ 4310N # 13N \$ 4310N		. 24		. 25	.27	-30	. 25	
* 011AA TAN	.21	.20	1 1	.19	.21	.20	11	
SSJU (ISW) SSJU (ISW)	225.0	225.0	225.0	225.0	225.0	225.0	225.0	
SSJATS (ISA)	46.2	44.6	1 1	43.6	48.2	46.1	1 1	
SSJW (ISW) WESS	46.2 66.5	44.6 53.9	53.7	43.6 57.2	48.2	46.1 68.5	57.2	
ONOS (SAIN) SSIATS (ISA)	30.7	29.9 36.1	35.3	29.3 38.4	31.7	31.4	38.1	
CANN (29) LOAD	6.66	5.50	5.37	5.83	6.10	7.06	5.77	stress
LEWETH CHACK  LEWETH CHACK  LINAL  LEWETH CHACK  LINAL  LEWETH CHACK  LE	4.63	4.55		4.45	4.80	4.75	1 1	nc/yield s/yield s
THICKNESS (IN.)  LENGTH CARCK	1.009	. 994	1,034	. 986	1,032	.963	1,005	ess at Pr re stres
WIOTH (IN.)	. 0502	. 0505	. 0505	. 0505	. 0503	. 0503	. 0502	Net section stress at $P_{nc}/y$ ield stress Nominal fracture stress/yield stress
NOITSTRION	3.00	3.02	3.01	3.01	3.01	3.01	3.01	Net se
370	Т	L	Ħ	Ľ	Ħ	T	T	* * * 2

Indicates occurrence of a distinct pop-in

TABLE 4. FRACTURE TEST RESULTS FOR 17-7 PH

N									
d0d 12h									
MI-404 LOWILSIO				!	-	+			
1211301	55.4	52.1	50.0	51.9	53.6				
(41) a,	. 014	. 012	.011	. 012	. 013	1.1			
* 01/10 N STAENOTAN			. 81	1 8 2	.80				
* 017 MET MATIO *	. 33	.31	.30	.31	.32				
SSJATS (ISN) SSJATS (ISN)	186.6	186.6	186.6	186.6	186.6	186.6			
134	62.2	58.6	56.2	58.1	60.4	1 !			
SSTA (18N) SSTA (18N) SSTA (18N)	62.2 147.9	58.6 144.6	56.2 151.7	58.1 153.6	60.4	147.7			
0 NO J (SA) SSOND (15N)	41.9	39.2 96.8	37.6 101.6	38.9 102.8	40.8	98.2			n
ONOL JANIJARA (SqlN)	10.70	10.38	10.80	11, 10	10.80	10.60	stress	tress	ct pop-i
CAIN)  OVOTONO  TENCH  CHOCK  (IN)  OVOTONO  TO T	4.50	4.20	4.00	4.20	4.35		1c/yield	s/yield s	a distin
THICKNESS (IN.)	. 980	. 991	. 992	666.	. 973	1,005	ess at P <sub>1</sub>	re stress	rence of
THICKNESS	. 0375	. 0358	. 0354	. 0358	. 0356	. 0360	Net section stress at $P_{ m nc}/{ m yield}$	Nominal fracture stress/yield	Indicates occurrence of a distinct pop-in
NOITOJAIO (NI) HIOIM	3,01	2.99	3.00	3, 02	3.00	3,00	Net sec	Nomina	Indicate
3810	I	Ы	Н	Н	I	н	<i>→</i>	* 2	4

TABLE 4. FRACTURE TEST RESULTS FOR 17-7 PH (cont.)

WIOTH (IN.)  THICKNESS	99     .0353     1.008     3.40      32.2     48.5     48.5     48.5     185.4     .26      .009     43.0         8.95     84.7     127.7      .69      .69	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	00 .0352 .981 4.00 37.9 56.4 56.4 185.4 .30012 50.0666666	.00 .0355 .988 3.95 37.1 55.4 55.4 185.4 .30011 49.2 † 8.27 77.8 116.06363	.00 .0348 .992 4.00 38.3 57.1 57.1 185.4 .31012 50.8 8.63 82.6 123.2666666	00 .0355 .981 3.90 36.6 9.30 87.3	.00 .0354 .927 4.10 8.90 83.8 121.4 65	.00 .0355 .920 4.20 39.5 57.0 57.0 185.4 .31012 50.2 9.36 88.0 126.96868	Net section stress at ${ m P}_{ m nc}/{ m yield}$ stress	Nominal fracture stress/yield stress	Indicates occurrence of a distinct non-in
(NI) HIOIM									ection str	nal fractu	ates occur
NOITOZAIO	T 2.99	T 3.00	T 3.00	T 3.00	T 3.00	T 3.00	T 3.00	T 3.00	1 Net se	** 2 Nomir	† Indica

TABLE 5. FRACTURE TEST RESULTS FOR AM 350

L 3.02 .0504 .995 6.88 45.2 67.5 67.5 170.1 .40020 60.8  L 3.02 .0504 .995 6.88 45.2 67.5 67.5 170.1 .4096  L 3.01 .0502 .999 6.60 16.60 110.0 166.496  L 3.02 .0490 .956 6.92 46.8 68.6 68.6 170.1 .4096  L 3.02 .0490 .956 6.92 46.8 68.6 170.1 .4096  L 3.02 .0490 stress at P <sub>nc</sub> /yield stress  † Indicates occurrence of a distinct pop-in									S.				H.			
3.02         .0504         .995         6.88          45.2         67.5         67.5         170.1         .40          .020         60.8           3.01         .0490         1.003         6.72          45.2         68.2         68.2         170.1         .40          .96            3.01         .0490         1.003         6.72          45.2         68.2         68.2         170.1         .40          .98           .98          .98          .98          .98          .98          .98          .98          .98          .98          .98          .98          .98          .98          .99          .99          .98          .98          .98          .99          .99          .99          .99          .99          .99          .99          .99          .99          .99	1	NOTABLE NOTABL	(W) HION	NI) SSANN	0	Sdly	18 SSOH9	SSJW (ISN)	17.	SSJW (ISW)		* 017 N N 10 N			UNISNI ISIO	NI- dOd 10NII
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	П	3.02	. 0504	. 995	6.88	16.60	45.2			170.1	.40	96.	.020		1	
3.00 .0502   1.019   6.60     43.6   65.2   65.2   170.1   .38     .019   58.6       16.65   110.0   164.6         .97             16.55   110.0   164.6         .97	T	3.01	. 0490	-	6.72					170.1	.40	86.	. 021			
3.00 .0503 1.019 6.79 44.9 68.0 68.0 170.1 .40021 61.3	I	3.01	. 0502	666.	6.60	9.	600	65.2 164.6	5.	170.1	.38		. 019		+	
3.02 3.02 3.02 3.02 6.92 46.8 68.6 68.6 170.1 40021 61.5  Net section stress at P <sub>nc</sub> /yield stress  Nominal fracture stress/yield stress  Indicates occurrence of a distinct pop-in	I	3,00	. 0503	1,019	6.79	, 65	44.9		68.0	170.1	.40	96.	. 021	1:1	1 !	
T 22 +	П	3.02	. 0490	. 956	6.92	17.00			1 . 1		.40	66.	. 021	1:1	Ti	
c7 +-		Net se	ection str	ress at P	nc/yield	stress									T	
† Indicates occurrence of a distinct pop-in			ıal fractu	ire stres	s/yield	stress										
	+	Indica	tes occu	rrence o	f a distir	et pop-ir	J									

TABLE 5. FRACTURE TEST RESULTS FOR AM 350 (cont.)

NJ-do											
NI-dOd LONILSIO	1			-	1						_
Anc (Ve.	! !	55.5	53.7	55.0	53.2	54.0 +	54.8	52.5			
(uj) o,	1 1	. 018	.017	. 018	.017	.015	.016	.014			
* 017A T3N * 017A T30N * 017A * 017A						06.		.91			
SSJAIS (ISN)		.38	.37	.37	.36	.35	.35	.33			
SSJULS W)	163.3	163.3	163.3	163.3	163.3	175.0	175.0	175.0			
SSIATS (ISN) SPAESS (ISN) SSIATS (ISN)		61.8	59.8	60.7	59.2	60.4	60.9	58.3			
SSZHIS(ISN)	139, 1	61.8 141.5	59.8 142.5	60.7 152.4	59.2 141.5	60.4 157.6	60.9	58.3 158.8			
OAOL (SAIM) OAOL (SAIM) SSIATS (ISM)	92.1	41.6	39.8 94.8	36.5	39.4 94.1	40.6 105.8	38.0 101.8	34.9 95.0			in
TO T	13,90	 14.30	14.30	13.80	 14.10	16.30	15.40	 14.70	d stress	stress	nct pop-
LEWETH CRACK	1 1	6.25	6.00	5.5	5.90	6.25	5.75	5.40	nc/yield	ss/yield	f a disti
THICKNESS (IN.)  INITIAL CAACK	1,020	. 989	1,009	1.20	1,010	. 989	1,125	1,206	ess at I	ire stre	rrence o
WIOTH (IN)	020	. 0498	. 0500	020	. 0496	.0512	. 0505	.0515	Net section stress at ${ m P}_{ m nc}/{ m yield}$	Nominal fracture stress/yield	Indicates occurrence of a distinct pop-in
WOITSTAID	3.02	3.02	3.02	3.01	3.02	3.01	3.00	3.00	Net se		Indica
310	T	T	T	T	T	T	H	T	*	* 2	+

TABLE 6. FRACTURE TEST RESULTS FOR VASCO JET 1000

SSARS STARS STARS STARS STARS SSARS (N) ASIA STARS SSARS (N) NOW (NSI) STARS SSARS (NSI) NOTON (NSI) NOTON STARS SSARS (NSI) NOTON (NSI) NOTON STARS SSARS (NSI) NOTON (NSI) N	250.6	3.45 19.5 28.7 28.7 250.6 .11002 25.1 4.07 23.0 33.81313	3.60 20.1 28.9 28.9 250.6 .12002 25.1 4.38 24.5 35.21	3.60 20.1 28.6 28.6 250.6 .11002 24.8 4.34 24.2 34.51414	3.50 19.7 29.0 29.0 250.6 .12002 25.5 + .56 25.7 37.8 + .1515 +	3.55 19.9 29.8 29.8 250.6 .12002 26.2 4.45 25.0 37.31515	yyield stress	Nominal fracture stress/yield stress	
ADJ 3AUN)	22.	19.	- 20. 38 24.	- 20. 34 24.	- 19. 56 25.	- 19. 45 25.	tress	ress	
LENGTH CAACH							Net section stress at P <sub>nc</sub> /yield s	ss/yield st	1
SSZWYDIHI	1.047	. 961	. 912	. 896	. 968	986 . 886	stress at	cture stre	
(NI) HIOIM	3.00 .0592	3.00 .0590	3.00 .0595	3.01 .0595	3.01 .0591	3.01 .0593	t section	ominal fra	
NOITOJAIO	L 3	L 3	L 3	L 3	L 3	L	* 1 Ne	** 2 No	-

TABLE 6. FRACTURE TEST RESULTS FOR VASCO JET 1000 (cont.)

NJ-do									
(U) TOU LONITSIO				. 1					_
(m) ISN) ou N	-	-		-!	+	-			
	1 1	23.3	26.3	23.9	25.5	26.2			
H19** * 011AA (n1) a'		. 002	. 002	. 002	. 002	. 002			
43/0	. 13	. 13	. 15	.13	.13	-14			
* 011NA TAN		.11	. 13	.12	.12	.13			
TIII	235.4	235.4	235.4	235.4	235.4	235.4			
131	11	26.6	30.2	27.5	29.0	29.8			
SSJUS (ISN) SSJUS (ISN) SSJUS (ISN)	29.7	26.6	30.2	27.5 30.9	29.0 31.6	29.8			
ONOS (SQL)	20.1	18.0 21.2	21.0	19.0	19.1 20.8	19.9 22.6			g
ONOL ONO ONOL ONO ONOL ONO ONOL ONO ONOLONO	3.70	3.90	4.64	3.93	3.82	4.15	stress	stress	ict pop-i
(in d	11	3.30	3.85	3.50	3.50	3.65	at P <sub>nc</sub> /yield	s/yield	a distir
THICKNESS (IN.)	. 973	976.	. 914	. 923	1,025	. 995	ess at P	re stres	rence of
THICKNE	. 0612	. 0612	. 0611	. 0612	. 0612	. 0611	Net section stress	Nominal fracture stress/yield st	Indicates occurrence of a distinct pop-in
NO1703A10 (N1) H101W	3.01	3.00	3.00	3.00	3.00	3.00	Net sec	Nomina	Indicat
3010	E	H	H	H	H	L	r⊢1 *	* 2	+

#### APPENDIX I

## COMPUTER PROGRAM

#### DISCUSSION

The computer program, described below, was written to expedite the reduction of fracture toughness data. Two paths of calculation are available as distinguished by the value of a control variable JJ. When JJ is set equal to 1 (one) and suitable compliance gage data are supplied, the computer calculates the critical crack length of the material before calculating  $K_{\text{C}}$  and its associated parameters. If the critical crack length of the material is known, the compliance calculation will be bypassed by setting JJ equal to 2. The data input format for JJ=1 and 2 is discussed below.

This program has been written in IBM 7094 Fortran II language. To provide a simplified means of data input, closed subroutines VDECOM and DECDCP have been included. When using these subroutines, the input parameters for the source program can be sequentially placed on data cards in Columns 1 to 70. However, at least one space must be left between each entry and no single entry may overlap to the following data card.

The critical crack length calculation path (JJ=1) has been programmed using the Naval Research Laboratory's procedure. A fifth degree polynomial curve fit has been used to obtain the following equations:

$$\frac{Ev}{\sigma W} = f\left(\frac{\pi a}{W}\right) \text{ and } \frac{\pi a}{W} = f\left(\frac{\sigma W}{Ev}\right)$$

These polynomial coefficients must be supplied as input data in the manner described below, regardless of the computation path desired. As written, the compliance calculation requires four sets of load-deflection values as input. The first three sets of values are read from the linear portion of the compliance curve. These values are averaged to obtain a representative value of  $E_{\rm V}/\sigma{\rm W}$  which determines the magnitude of the calibration curve shift. The fourth set is the load-deflection values at fracture.

The basic fracture toughness calculations are made using Irwin's tangent equation

$$K^2 = \sigma_g^2 W \tan \left( \frac{\pi a_o}{W} \right)$$

The corrected K values incorporate the plastic zone correction factor where

$$a = a_0 + r_p$$
 and  $K^2 = \sigma_g^2$  W tan  $\left(\frac{\pi a_0}{W} + \frac{K^2}{2W\sigma_{ys}^2}\right)$ 

When the argument of the tangent function exceeds the mathematical limitations of Irwin's equation, no corrected K can be calculated, thus the same value as K basic is printed in the data output.

When all the input parameters necessary to make fracture toughness calculations are known (JJ=2), the source program will bypass the compliance calculation of the critical crack length.

An example of the data output from this computer program is shown in Table 7.

## DATA INPUT

All input data, except JJ, must be decimal values. The first two data cards for both JJ=1 and JJ=2 contain the polynominal curve fit constants shown below. The minimum one space format using columns 1 to 70 apply to all the data card input discussions. The constant data cards numbers 1 and 2 are supplied only once for all program runs.

#### Card No. 1

.31989 .22831 .15115 0. 0. .38496

#### Card No. 2

1.23184 -.212053 -.0774244 0. .0019483

JJ=1: Actual test data is read in on subsequent cards, beginning with card number 3. Each set of test data is begun in column one and may be presented on several successive cards, as necessary. A value must be given for every parameter listed on the card below. If applicable, 0.0 may be inserted as the value for any of these items.

Sample Data Card for JJ=1

$$-JJ$$
 $-A\phi$ 
 $-A\phi$ 
 $-A\phi$ 
 $-A\phi$ 
 $-B$ 
 $-B$ 

NOTE: This data card yields the first two lines of results in Table 7.

JJ=2: For calculations not requiring the compliance calculation of the critical crack length, the following sample data card is appropriate. The one space between data items in Columns 1 to 70 is required.

Sample Data Card for JJ=2

The following is an abbreviated glossary for the symbols shown on the sample data cards. No attempt has been made to define all the symbols used in the source program and the subroutines VDECOM and DECDCP.

## LIST OF SYMBOLS (INPUT DATA)

Fortran Symbol	Analytical Symbol	Definition
$Aoldsymbol{\phi}$	$2a_{o}$	Total initial crack length
W	w	Width
В		Thickness
PIC		"Pop-in" load (Kips)
SY		Yield Stress (KSI)
FAC		Total magnification of the compliance gage*
${f E}$	E	Modulus of Elasticity (psi)
P(1), P(2), P(3)		Loads (Kips) on linear portion of compliance curve
DEF(1), DEF(2), DEF(3)		Deflections (inches of chart) corresponding to P(1), P(2), P(3)
P(4) or PF		Load at Fracture (Kips)
DEF (4)		Deflection at fracture (inches of chart)
A	2a	Total critical crack length

<sup>\*</sup> A gage with a 250X microformer and a gage lever ratio of 2 would have FAC = 500.

# DATA OUTPUT

The computer program supplies a data output as shown in Table 7. The first line of a single test set contains  $K_{Ic}^{\rm data}$  while the second line contains  $K_{C}^{\rm data}$ . The ''nominal ratio'' as used here is equivalent to the ''notch strength ratio.''

# SOURCE PROGRAM AND SUBROUTINES

The complete Fortran II program which consists of the source program and two closed subroutines, VDECOM and DECDCP, is presented in Table 8.

TABLE 7. EXAMPLE OF COMPUTER PROGRAM OUTPUT

NESS K CORRECTED	69.69	61.96	53.04	26.40	28.17	26.84	39.24	69.14	49.84
TOUGHNESS BASIC CORP	66.66	59.65	51.63	26.24	28.01	26.69	38.76	66.25	48.72
PLAST IC ZONE	0.0236	0.0186	0.0146	0.0029	0.0029	0.0027	0.0064	0.0232	0.0121
LENGTH/ WIDTH	0.31	0.28	0.33	0.29	0.30	0.27	0.31	0.33	0.32
NET RATIO	0.52	0.48	0.42	0.19	0.19	0.19	0.28	0.52	0.38
NOMINAL	66.0	96*0	1.01	0.21	0.21	0.21	0.33	1.01	1.02
YIELD STRESS	181.00	181.00	175.00	37.64 197.00 42.75 197.00	39.99 208.00 45.37 208.00	38.93 206.00 45.42 206.00	196.00 196.00	181.00	181.00
STRESS	94.81	86.23 181.00 255.72 181.00	72.91 175.00 261.28 175.00	37.64	39.99	38.93	55.22	93.62 181.00 249.78 181.00	69.11 181.00 265.38 181.00
NOMINAL	94.81 179.38	86.23	72.91 177.06	37.64	39.99	38.93	55.22	93.62	69.11
GROSS	11.10 65.15	10.80 62.33 21.45 123.79	49.16 119.39	26.60	27.85	28.51	38.13	63.09	47.09
LOAD	11.10	10.80	8.40	4.60 5.08	4.70	4.96	6.50	10.80	8.10
WICTH THICKNESS	0.0871	0.0880	0.0870	0.0882	0.0862	0.0888	0.0872	0.0873	0.0879
WICTH	1.956	1.969	1.964	1.961	1.958	1.959	1.955	1.961	1.957
FINAL	1.066	1.016	1.067	0.614	0.621	0.568	0.662	966.0	1.031
CRACK LENGTH INITIAL FINAL	0.612	0.546	0.640	0.575	0.594	0.524	0.605	0.640	0.624

#### TABLE 8. SOURCE COMPUTER PROGRAM

```
CCALC CENTER NOTCH FRACTURE TOUGHNESS AND COMPLIANCE GAGE CALCULATIONS
      DIMENSION AFIT(7), BFIT(7), P(4), V(3), DEF(4), Y(3), INT(10),
     1 DEC(10)
      PI = 3.14159
      JTAPF = 2
      NTAPF = 3
C
       READ IN COEFFICIENTS FOR COMPLIANCE CURVES
        Y = F(X)
      READ INPUT TAPE JTAPE, 1000, (AFIT(I), I = 1, 7)
C
        X = F(1 \cdot /Y)
      READ INPUT TAPE JTAPE, 1000, (BFIT(I), I = 1, 7)
      WRITE OUTPUT TAPE NTAPE, 4000
    5 N = 1
      KPASS = 1
      CALL VDECOM( N. INT. DEC. KPASS)
      JJ = INT(1)
      N = 7
      KPASS = 3
      CALL VDECOM( N. INT, DEC. KPASS)
      \Delta \Omega = DFC(1)
      AO = AO
      W = DFC(2)
      B = DFC(3)
      PIC = DEC(4)
      SY = DEC(5)
      FAC = DFC(6)
      F = DFC(7)
      A = DEC(6)
      PF = DEC(7)
      NN = 0
      GO TO(15,35),JJ
   15 N = 8
      KPASS = 3
      CALL VDECOM( N, INT, DEC, KPASS)
      00 \ 10 \ I = 1, 4
      K = 2 * I
      P(I) = 1000 \cdot *DEC(K-1)
   10 DEF(I) = DEC(K)
C
       COMPUTE AVERAGE VALUES OF Y AT Z
      YSUM= 0.
      00 20 I = 1, 3
      V(I) = 0.5 * DEF(I) / FAC
      Y(I) = E * V(I) * B / P(I)
   20 YSUM=YSUM + Y(I)
      YAVG=YSUM/3.
      YFRAC=F*B*DFF(4)/(2.*FAC*P(4))
      7 = PT * AO / (2. * W)
       FVALUATE COMPLIANCE POLYNOMIAL AT Z
(
      YFIT = AFIT(1)
      DO 25 I = 1, 6
   25 YFIT = YFIT + AFIT(I+1) * Z**I
       COMPUTE DIFFFRENCE BETWEEN Y VALUES
(
      YDIFR = YAVG - YFIT
       CALCULATE ZNEW AT FRACTURE FROM ADJUSTED VALUE OF Y
(
      YNEW=YERAC - YDIFR
```

END

```
TABLE 8. (Continued)
      7NEW = BFIT(1)
      DO 30 I = 1, 6
   30 7NEW = ZNEW + BFIT (I+1) / (YNEW**I)
       COMPUTE NEW FRACTURE CRACK LENGTH
C
      A = 2. * W * 7NEW / PI
      PF=P(4)/1000.
       COMPUTE THE STRESS AND FRACTURE-TOUGHNESS
(
   35 AKIC = 0.0
      SG = PIC/(W*B)
      SNF = PIC/(B*(W-AO))
      SN1 = SNF
      SN2 = PIC/(B*(W-A1))
      SRN = SN2/SY
      PA = PI*AO/(2.*W)
      BKS = SINF(PA)/COSF(PA)*W*SG**2
      AOW = AO/W
      SR = SNF/SY
      RP = BKS/(2 \cdot *PI *SY **2)
      BK1 = SQRTF(BKS)
(
       EVALUATE THE EQUATION LIMITATION
      GOYS = (SG/SY)**2
      QM = SQRTF(2 \cdot /GOYS - 1 \cdot )
      PAM = ATANF(QM) - QM*GOYS/2.
      IF (PAM-PA) 39,40,40
   39 AKIC = BK1
      GO TO 45
       COMPUTE THE PLASTIC ZONE FRACTURE-TOUGHNESS
(
   40 PAP = PA + BKS/(2.0*W*SY**2)
      PKS = SINF(PAP)/COSF(PAP)*W*SG**2
      BK = SQRTF(BKS)
      AKIC = SQRTF(PKS)
      BKS = PKS
      IF (AKIC-BK-.005)44,40,40
   44 RP = PKS/(2.*PI*SY**2)
   45 IF (NN) 50,50,60
   50 WRITE OUTPUT TAPE NTAPE, 2000, AO, W, B, PIC, SG, SN1, SNF, SY, SR, AOW, RP,
     1 BK1, AKIC
      \Delta 1 = \Delta 0
       AO = A
      PIC = PF
      NN = 1
       REPEAT FOR PLANE-STRESS TOUGHNESS
(
      IF(AO)5,5,35
   60 WRITE OUTPUT TAPE NTAPE, 3000, A1, A0, W, B, PIC, SG, SN2, SNF, SY, SRN, SR,
     1 RP, BK1, AKIC
      GO TO 5
 1000 FORMAT(
                7F10.5)
 2000 FORMAT(/,F7.3,F15.3,F9.4,F8.2,F7.2,F9.2,F8.2,F7.2,F14.2,F8.2,F9.4,
      1
       2F9.21
 3000 FORMAT( F7.3, F8.3, F7.3, F9.4, F8.2, F7.2, F9.2, F8.2, 3F7.2, F1.7.4,
         2F9.21
      1
 4000 FORMAT(1H1,116H CRACK LENGTH WIDTH THICKNESS LOAD GROSS
                                                                          NOM
                                                                           K,
                   YIELD NOMINAL NET LENGTH/
                                                     PLASTIC
                                                                TOUGHNESS
      1 I NAL
            NET
                                                                          STR
                                                         STRESS
                                                                  STRESS
      2/,120H INITIAL FINAL
                                                      BASIC CORRECTED)
      3ESS STRESS RATIO RATIO
                                   WIDTH
                                             ZONE
```

```
, KPASS)
                                                        12/18/63
                               KARRAY, P
     SUBROUTINE VDECOM(N ,
                                KARRAY, P
                                             , KPASS)
     SUBROUTINE VDECOM(N ,
     DIMENSION P(10)
     DIMENSION KINPUT(72), KARRAY(10), PARRAY(10), AINPUT(72),
                KCUTPT(72), ACUTPT(72)
     EQUIVALENCE (PLUS, IPLUS), (AMINUS, IMINUS), (DECPT, IDECPT),
                  (COMMA, ICOMMA), (E, IE), ( BLANK, IBLANK),
     1
                  (AINPUT, KINPUT), (ACLTPT, KOUTPT)
      KPASS = KPASS
       J2 = INPLT TAPE
C
      J2 = 2
      J4 = SCRATCH TAPE
C
      J4 = 4
      GC TO ( 2, 2, 2, 3), KPASS
    2 REWIND J4
    3 NUMCCP = N
      NEX = 1
      N1 = 1
      L = 1
      GO TO (1, 10, 50, 50), KPASS
    1 CCNTINUE
       SET UP CHARACTERS FOR LATER TEST
C
            = 206060606060
      PLUS
B
      AMINUS = 406060606060
B
      DECPT = 336060606060
B
      CCMMA = 736060606060
В
             = 256060606060
В
       E
      BLANK = 606060606060
B
       READ ALPHANUMERIC CHARACTERS
C
     5 READ INPUT TAPEJ2,1000, (AINPUT(J), J=1,72)
       I = 1
       GC TO (10,10, 50, 50), KPASS
       DECOMPOSITION OF INTEGERS
C
    10 DO 21 N = N1, NUMBCP
       N1 = N
        SEARCH FCR START OF NUMBER
   101 IF (KINPUT(I) - IBLANK) 102, 11, 102
   1C2 IF (KINPUT(I) - ICOMMA) 12 , 11, 12
    11 I= I+1
       IF( I -72)1C1,1C1, 5
        SELECT INTEGERS
    12 L1 = L
       M = 0
       DC 20 J=1,6
       KOUTPT(L) = KINPUT(I)
       IF( KINPUT(I) - IMINUS) 122, 120, 122
   122 IF( KINPUT(I) - IPLUS) 123, 120, 123
   120 M = 1
   123 L = L + 1
       I = I + 1
       IF(KINPUT(I) - IBLANK) 121 ,13 ,121
   121 IF(KINPUT(I) - ICOMMA) 20 ,13 , 20
        RIGHT ADJUST IN FIELD
    13 IF( J-6) 14,21,14
     14 KCO = J - M
       DC 15 K = 1, KDC
        L2 = L1 + 6 - K
```

```
(KPASS) 12/18/63
                                KARRAY, P
      SUBROUTINE VDECOMIN ,
   L3 = L1 + J - K
15 KCUTPT( L2 )= KCUTPT( L3 )
      L = L1 + 6
      KDO = 6-J + M
      KGO = 1 + M
      DC 16 K = KGO \cdot KDC
      L4 = L1 + K
                    -1
   16 \text{ KGUTPT(} \text{ L4)} = 0
      GC TO 21
   20 CCNTINUE
   21 CONTINUE
      IEND = 6* NUMDCP
       WRITE ALPHANUMERIC CHARACTERS
C
      WRITE OUTPUT TAPEJ4,1000, (AOUTPT(J), J=1, IEND)
      REWINCJ4
       READ INTEGER LIST
C
      READ INPUT TAPEJ4,1001, (KARRAY(J), J=1, NUMDCP)
      REWINDJ4
   40 RETURN
   50 CALL DECDCP ( IPLUS, IMINUS, IDECPT, ICOMMA, IE, IBLANK,
     1 KINPUT, NUMDCP, PARRAY, I, L, NEX, N1, KPASS)
      NEX = NEX
      GO TO(30,5), NEX
   30 DC 31 J = 1, NUMDCP
   31 P(J) = PARRAY(J)
      GC TO 40
 1000 FCRMAT(72A1)
 1001 FCRMAT(1116)
      END(1,1,0,0,0,1,1,1,0,1,0,0,0,0,0,0)
```

```
SUBROUTINE DECDCP( IPLUS, IMINUS, IDECPT, ICOMMA, IE, IBLANK, 12/18/63
      SUBROUTINE DECDCP( IPLUS, IMINUS, IDECPT, ICOMMA, IE, IBLANK,
     1 KINPUT, NUMDCP, PARRAY, I, L, NEX, N1, KPASS)
      EQUIVALENCE (AOUTPT, KOUTPT)
      DIMENSION AOUTPT(72), KOUTPT(72), PARRAY(10), KINPUT(72)
      NEX = NEX
       J2 = INPUT TAPE
C
      J2 = 2
C
       J4 = SCRATCH TAPE
      J4 = 4
      M = L
      GC TO (50,51), NEX
       DECOMPOSITION OF DECIMAL AND EXPCNENTIAL NUMBERS
C
       LIMIT DECOMPOSITION TO 6 NUMBERS
   50 IF(NUMDCP -6)503,503,502
  5C2 IDEC = 6
      IEND = 12*ICEC
      GC TO 504
  503 IDEC = NUMDCP
      IEND = 12# NUMDCP
  504 IF(I -72) 51,51,505
      NEX = 2
  505
       GO TC 30C
   51 \text{ NEX} = 1
      DO 100 N= N1, IDEC
      N1 = N
       SEARCH FOR START OF NUMBER
  510 IF(KINPUT(I)- IBLANK) 52, 53, 52
   52 IF(KINPUT(I)- ICOMMA) 54, 53, 54
   53 I= I + 1
      IF( I - 72) 510, 510, 505
   54 M1 = M
       STORE NUMBERS UP TO DECIMAL POINT
  541 IF(KINPUT(I) - IDECPT) 55, 65, 55
   55 KCUTPT(M) = KINPUT(I)
      I = I + 1
      M = M + 1
      GC TO 541
       TEST FOR END OF NUMBER OR EXPCNENTIAL
C
   60 IF(KINPUT(I) - IE ) 61, 70,61
   61 IF(KINPUT(I) - IPLUS ) 62, 70,62
   62 IF(KINPUT(I) - IMINUS ) 63, 70,63
   63 IF(KINPUT(I) - ICCMMA ) 64, 80,64
   64 IF(KINPUT(I) - IBLANK ) 65, 80,65
       STORE CECIMAL POINT AND NUMBERS
   65 \text{ KCUTPT(M)} = \text{KINPUT(I)}
      I = I + 1
      M = M + 1
      GC TO 60
       COMPLETE EXPONENTIAL FIELD THROUGH 8 LCCATIONS
   70 \text{ LDO} = \text{M1} + 7
      DC 71 J1= M, LDO
   71 \text{ KCUTPT}(J1) = 0
      M = M1 + 8
       STORE E IN LCCATION 9
      KCUTPT(M) = IE
      IF(KINPUT(I)- IE) 73,72,73
```

SUBROUTINE DECDCP( IPLUS, IMINUS, ICECPT, ICOMMA, IE, IBLANK, 12/18/63 72 I= I+1 C TEST FOR SIGN OF EXPONENT 73 IF(KINPUT(I)-IMINUS) 74, 76,74 74 IF(KINPUT(I)-IPLUS ) 75, 76 ,75 75 KOUTPT(M+1) = IPLUSGC TO 77 C STORE SIGN 76 KCUTPT(M+1) = KINPUT(I)I = I + 1TEST FCR END OF EXPONENT 77 IF(KINPUT(I+1) - IBLANK) 78, 79, 78 78 IF(KINPUT(I+1) - ICOMMA) 791,79 ,791 79 KCUTPT(M+2) = 0 KCUTPT(M+3) = KINPUT(I)GC TO 792 791 KCUTPT(M+2) = KINPUT(I)KOUTPT(M+3) = KINPUT(I+1)792 I= I+2 M = M1 + 12GC TO 100 C COMPLETE DECIMAL FIELD 80 LCO = M1 + 11DO 81 J1 = M , LDO 81 KCUTPT(J1) = 0M = M1 + 121CO CONTINUE WRITE ALPHANUMERIC CHARACTERS WRITE OUTPUT TAPEJ4, 1000, (AOUTPT(J), J=1, IEND) IF(NUMCCP - ICEC ) 201, 201, 200 200 M = 1N1 = N1 + 1IEND = 12\*(NUMDCP - 6) IDEC = NUMBCP GC TO 504 201 GC TU( 202, 202, 202, 300), KPASS 2C2 REWINDJ4 READ DECIMAL AND EXPONENTIAL LIST READ INPUT TAPE J4,1010, (PARRAY(J), J=1, NUMDCP) REWINDJ4 300 L = MRETURN 1000 FCRMAT(72A1) 1010 FORMAT(6F12.5) END(1,1,0,0,0,1,1,1,0,1,0,0,0,0,0)

### APPENDIX II

# COMPLIANCE METHOD ANALYSIS

From the theory of elasticity as presented by Timoshenko and Goodier (Reference 7), there is a stress function  $\phi$  which satisfies the biharmonic equation  $\Delta^4 \phi$ =0 and also satisfies boundary conditions such that the external forces may be considered as an extension of the internal stress distribution. If this is true then:

$$\sigma_{x} = \frac{\partial^{2} \phi}{\partial y^{2}}$$
,  $\sigma_{y} = \frac{\partial^{2} \phi}{\partial x^{2}}$  and  $\tau_{xy} = -\frac{\partial^{2} \phi}{\partial x \partial y}$ 

where  $\phi$  is a function of x and y and is called the Airy stress function.

Westergaard (Reference 8) has provided a two-dimensional stress analysis of a very large flat plate with tension applied in the y direction and a system of cracks along the x axis each of length 2a and center at  $x=0,\pm L,\pm 2L\cdots$  Using Westergaard's notation, if the function  $\phi$  is related to a complex function, we shall call it Z where

$$Z = Z(z) = Z(x+iy) Re Z + i Im Z$$
 (I)

and the pertinent values are given by the analytic function

$$Z = \frac{\sigma}{\sqrt{1 - \left(\frac{\sin(\frac{\pi \sigma}{L})}{\sin(\frac{\pi z}{L})}\right)^2}}$$
 (2)

In Westergaard's notation,

$$Z' = \frac{dZ}{dz}$$

$$Z = \frac{dZ}{dz}$$

$$\overline{Z} = \frac{d\overline{\overline{Z}}}{dz}$$

Applying the Cauchy Riemann conditions,

$$\frac{\partial \operatorname{Re} Z}{\partial x} = \frac{\partial \operatorname{Im} Z}{\partial y} = \operatorname{Re} Z', \text{ and } \frac{\partial \operatorname{Im} Z}{\partial x} = -\frac{\partial \operatorname{Re} Z}{\partial y} = \operatorname{Im} Z'$$

If the Airy function is defined as

$$\phi = \text{Re } \overline{\overline{Z}} + y \text{ Im } \overline{Z}$$

and it satisfies the biharmonic equation

$$\frac{\partial^4 \phi}{\partial x^4} + \frac{2 \partial^4 \phi}{\partial x^2 \partial y^2} + \frac{\partial^4 \phi}{\partial y^4} = 0$$

then

$$\sigma_{y} = \frac{\partial^{2} \phi}{\partial x^{2}} = \frac{\partial^{2} \operatorname{Re} \overline{\overline{Z}}}{\partial x^{2}} + y \frac{\partial^{2} \operatorname{Im} \overline{Z}}{\partial x^{2}}$$

Since

$$\frac{\partial^2 \operatorname{Re} \overline{\overline{Z}}}{\partial x^2} = \operatorname{Re} Z$$

and

$$y \frac{\partial^2 \operatorname{Im} \overline{\overline{Z}}}{\partial x^2} = y \operatorname{Im} z'$$

Therefore,

$$\sigma_{v} = Re Z + y Im Z'$$
 (3)

By similar steps,

$$\sigma_{x} = \text{Re } Z - y \text{ Im } Z'$$
 (4)

$$\tau_{xy} = -y \operatorname{Re} Z'$$
 (5)

The displacement, v, in the y direction for a plane strain situation is given by the equation.

$$Ev = 2 \left( 1 - \nu^2 \right) Im \overline{Z} - \left( 1 + \nu \right) y Re Z$$

where  $\nu$  is Poisson's ratio and E is the modulus of elasticity.

It has been shown by Irwin (Reference 9) that the above stress analysis is a good approximation to the stress state in a centrally cracked sheet specimen whose vertical and horizontal axes of symmetry are taken as the yand x axes respectively. The specimen, with width w = L, is regarded as one unit of the crack system, and therefore the specimen edges are at x =  $\pm L/2$  and the crack extends from x = -a to x = +a. Boundary conditions require the stresses  $\sigma_x$  and  $\sigma_x$  to be zero along the borders of the crack and the stresses  $\sigma_x$  and  $\sigma_x$  to be zero along the side boundries. All these conditions are fulfilled except the condition that  $\sigma_x$  = 0. Irwin remedied this by rewriting  $\sigma_x$  as

$$\sigma_{x} = \text{Re Z -y Im Z}' - \sigma_{0x}$$
 (6)

where  $\sigma_{o_{_{\mathbf{X}}}}$  is a constant stress adjusted so that

$$\int_{0}^{\infty} \sigma_{\chi} dy = 0 \text{ at } x = \pm \frac{L}{2} .$$

The constant  $\sigma_{o_X}$  does not make  $\sigma_{x}$  zero at all points on the side boundries, but it does reduce  $\sigma_{x}$  to a very small value. It is assumed here that this deviation from an exact solution results in errors much smaller than those resulting from the departure from linear elasticity theory for finite strains.

Irwin modified the equation for the displacement in the y direction to make it applicable to a generalized plane stress condition. This equation is

$$Ev = 2 \text{ Im } \overline{Z} - (1 + \nu) \text{ v Re } Z \tag{7}$$

It is now possible to derive an expression for compliance at the specimen edges. The edges of the specimen are at  $x = \pm L/2$ . Therefore,  $z = \pm L/2 + iy$  where y is 1/2 of the specimen gage length

Integration of  $\overline{Z} = \int Z dz$  utilizing Equation (2) gives

$$\overline{Z} = -\frac{L \sigma}{\pi} \sin^{-1} \left( \frac{\cos \frac{\pi z}{L}}{\cos \frac{\pi a}{L}} \right) + C$$
 (8)

where C is a complex constant of integration to be evaluated for the known conditions at a = o. Utilizing the identities

$$\cos \left[ \frac{\pi}{L} \left( \frac{L}{2} + iy \right) \right] = -i \sinh \frac{\pi y}{L}$$

$$\sin \left[ \frac{\pi}{L} \left( \frac{L}{2} + iy \right) \right] = \cosh \frac{\pi y}{L}$$

and

$$-\sin^{-1}(-m) = i \sinh^{-1}(m)$$

and substituting in Equation 8, it is found that

$$\overline{Z} = \frac{i L \sigma}{\pi} \sinh^{-1} \left[ \frac{\sinh \frac{\pi y}{L}}{\cos \frac{\pi a}{L}} \right] + \text{Re } C + i \text{ Im } C$$
 (9)

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Therefore,

$$\operatorname{Im} \overline{Z} = \frac{L \sigma}{\pi} \sinh^{-1} \left[ \frac{\sinh \frac{\pi y}{L}}{\cos \frac{\pi a}{L}} \right] + \operatorname{Im} C \tag{10}$$

Also, when  $x = \pm L/2$ ,

$$Z = \frac{\sigma}{\sqrt{1 - \left(\frac{\sin\frac{\pi a}{L}}{\cosh\frac{\pi y}{L}}\right)^2}}$$
(11)

and it is real. Therefore, Re Z = Z at x =  $\pm L/2$ . The expression for the displacement v is found by substituting Equations 10 and 11 into Equation 7. In accordance with R. W. Boyle (Reference 2), the compliance  $v/\sigma$ , is expressed as a dimensionless parameter by dividing by the specimen width w and multiplying by Young's modulus. Now, by setting L = w, the equation for compliance becomes

$$\frac{\text{Ev}}{\sigma w} = \frac{2}{\pi} \sinh^{-1} \left( \frac{\sinh \frac{\pi y}{w}}{\cos \frac{\pi a}{w}} \right) + \frac{2 \text{ImC}}{\sigma w} - \frac{y}{w} \sqrt{1 - \frac{\sin^2 \frac{\pi a}{w}}{\cosh^2 \frac{\pi y}{w}}}$$

The term Im C is evaluated by knowing that when a = o the specimen is simply an unnotched specimen, and  $E_V/\sigma$  W is equal to y/w. The final equation

$$\frac{Ev}{\sigma w} = \frac{2}{\pi} \sinh^{-1} \left( \frac{\sinh \frac{\pi y}{w}}{\cos \frac{\pi a}{w}} \right) - \frac{y}{w} \frac{(1+v)}{\sqrt{1 - \frac{\sin^2 \frac{\pi a}{w}}{\cosh^2 \frac{\pi y}{w}}}} + \frac{v.y}{w}$$
 (12)

To review the pertinent symbols in Equation 12,  $\underline{E}$  is Young's modulus,  $\underline{v}$  is one-half the specimen extension,  $\underline{\sigma}$  is the gross stress calculated from the dimensions of the uncracked specimen,  $\underline{w}$  is the specimen width,  $\underline{a}$  is one-half the crack length, and  $\underline{y}$  is one-half the specimen gage length.

It can be seen from Equation 12 that a plot of the dimensionless parameters  $E v / \sigma W$  vs  $\frac{\pi}{W}$  will yield a curve whose shape is constant regardless of the material tested. This is true because the factor y/W is constant for identical specimen geometries. The curve will shift up or down the  $E v / \sigma W$  axis for different materials because of different E values and the corresponding change in the extension factor vy/W.

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DOCUMENT					
(Security classification of title, body of abstract and indexi	NTROL DATA - R&D	ered when t	the overall report is classified)		
1. ORIGINATING ACTIVITY (Corporate author)	28. REPORT SECURITY CLASSIFICATION				
Air Force Materials Laboratory, Research a					
Technology Division, Air Force Systems Command,			2 b. GROUP		
Wright-Patterson Air Force Base, Ohio					
3. REPORT TITLE CENTER NOTCH PLANE STRAIN K <sub>IC</sub> FRAC	CTURE TOUGHNI	ESS PRO	OPERTIES OF SEVERAL		
HIGH-STRENGTH STEEL ALLOYS					
4. DESCRIPTIVE NOTES (Type of report and inclusive dates)					
January 1964-January 1965					
5. AUTHOR(S) (Last name, first name, initial) Davis, Sidney O., Tupper, Nathan G., 1/Lt,	USAF, Lagrone, I	Dana C	1/I.t. USAF.		
Niemi, Roger M.	Obni, Lagiono, 1	Dana C.	, 1/ 11, 00111 ,		
1110111, 110501 111.					
6. REPORT DATE	78. TOTAL NO. OF PAG	GES	7b. NO. OF REFS		
August 1965	42		9		
8a. CONTRACT OR GRANT NO.	9a. ORIGINATOR'S REPORT NUMBE				
E001	A FIRST MD 05 014				
b. PROJECT NO. 7381	AFML-TR-65-214				
Magla Na 799106					
c. Task No. 738106	9b. OTHER REPORT NO(S) (Any other numbers that may be assigned this report)				
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11. SUPPLEMENTARY NOTES	12. SPONSORING MILITARY ACTIVITY				
	Air Force Materials Laboratory, Research and				
	Technology Division, Air Force Systems				
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13. ABSTRACT					
Plane strain fracture toughness and tens	ile properties we	re dete	rmined at room tem-		
perature utilizing compliance and pop-in n	notheds for four	high at	monoth wheel allows		

ipliance and pop-in methods, for four high-strength steel alloys: PH 15-7 Mo, 17-7 PH, AM 350 and Vasco Jet 1000 (H-11). Fracture toughness values varied over a fairly wide range, with AM 350 having the highest at approximately 60.7 KSI in. and Vasco Jet 1000 (H-11) having the lowest at approximately 25 KSI in. A computer program used to reduce fracture toughness data was able to calculate critical crack length as well as fracture toughness when given either suitable compliance gage data or the measured test data. An acoustical pickup, used as an additional test monitor, is described. Analytical basis for the compliance method is presented.

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